



Engineering interfacial charge transfer channel for efficient photocatalytic H₂ evolution: The interplay of CoP_x and Ca²⁺ dopant

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ABSTRACT

Cobalt phosphide (CoP_x) has been developed as a cost-effective cocatalyst for photocatalytic H₂ evolution with the advantages of excellent conductivity, strong reduction ability, good thermal and chemical stability. In this work, a facile preparation strategy was proposed to fabricate ultrafine CoP_x nanoparticles on the surface of CdS. The effective H₂ production over ultrafine CoP_x nanoparticles has been realized. In addition, we found Ca²⁺ as an alkaline earth metal ion can promote the interaction between CdS and CoP_x. Both experimental results and density function theory indicate that the Ca²⁺ dopant can act as surface trapping sites on CdS and lead to efficient separation of photogenerated electron-hole pairs. The integration of CoP_x and Ca²⁺ dopant can synergistically enhance both photogenerated electron-hole separation as well as interfacial charge transfer, which enables a remarkable improvement on the H₂ generation performance of CdS. The photocatalytic H₂ generation rate of Ca-modified CoP_x@CdS can reach up to 2441.5 μmol h⁻¹ under optimal conditions with the apparent quantum efficiency as high as 35.4% at 420 nm. This finding motivates the development of simplified fabrication procedures for constructing and modifying cobalt active sites with efficient photocatalytic H₂ generation performance.

1. Introduction

Hydrogen (H₂) is known as a clean and sustainable energy carrier that plays an essential role in the low-carbon economic system. In the past decades, the developments of hydrogen energy system as well as applications of hydrogen in different fields have attracted worldwide interests [1–5]. Until now, a variety of photocatalysts have been exploited in the H₂ evolution reaction. However, semiconductor-based photocatalysts usually suffer from rapid electron-hole recombination, lack of reaction sites, and low efficiency in solar to chemical energy conversion. To overcome the aforementioned difficulties, a cocatalyst is usually required to enhance the separation of photogenerated electron-hole pair and to provide sufficient active sites for surface chemical reactions [6–13]. Generally, most highly efficient

photocatalytic systems are realized by utilizing noble-metal-based cocatalysts such as Pt, Pd, Ag, etc [14]. Unfortunately, noble metals are too scarce and expensive to be used for large-scale energy production. Therefore, the exploration and development of cost-effective cocatalysts are highly desired.

Of particular interest, cobalt-based transition metal phosphides have attracted worldwide attention as promising noble metal-free alternatives for noble-metal-based cocatalysts in photocatalytic H₂ production [15]. Amongst various cobalt-based materials, cobalt phosphides (CoP_x) including Co₂P, CoP, CoP₂, CoP₃, etc. have become the hot topic in catalytic H₂ production owing to the metal-like characteristics of Co–P bond [16–21]. With lower Fermi energy level, it was anticipated that CoP_x can play a significant role in promoting both electrocatalytic and photocatalytic H₂ evolution [22]. Previous studies indicated that

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rational construction of Schottky junction between CoP_x and semiconductor CdS could enable promising improvements on photocatalytic H_2 evolution performance [23–27]. Meanwhile, the efficient electron transfer property endows photogenerated electrons with great capability for reducing adsorbed H^+ into H_2 .

Moreover, it has been reported that metal cations can be employed to tailor the electronic structure of CdS for further enhancing the photocatalytic performance [28]. By doping with a donor impurity, the impurity levels below the conduction band may induce the generation of electron trap sites on the surface of CdS, which is believed to hinder photogenerated electron-hole recombination and increase the lifetime of photogenerated electrons [29–32]. Common alkaline-earth metal ions (such as Mg^{2+} , Ca^{2+} , Sr^{2+} , Ba^{2+}) with good stability and charge transition properties have been applied for modifying semiconductor [33]. Benefitted from the good electron-donating feature and stability of alkaline-earth metal ions, significant enhancement on the photocatalytic activities have been realized after doping with alkaline-earth metal ions [34]. However, it has rarely been applied in modifying CdS-based photocatalysts.

Herein, we proposed a convenient preparation strategy to fabricate ultrafine CoP_x nanoparticles and applied it into photocatalytic H_2 evolution. By utilizing organophosphorus compound triphenylphosphine (PPh_3) as phosphorus source, the as-fabricated CoP_x nanoparticles are intrinsically encapsulated by carbon layer and applied as the cocatalyst for photocatalytic H_2 evolution. To our notice, this is the first report about the design of carbon assisted CoP_x @CdS interfacial structure through the assistance of PPh_3 , which does not require additional hybrid carbon source via multi-step reactions. The relation between metal dopants and the catalytic performance has been investigated. Benefitted from the interfacial engineering, the charge transfer behavior between CdS and CoP_x has been modified through Ca^{2+} dopant. Specially, the photocatalytic H_2 generation over Ca-modified CoP_x @CdS can reach up to $2441.5 \mu\text{mol h}^{-1}$ with the apparent quantum efficiency of 35.4% at 420 nm. Finally, the mechanism was explored to reveal the effect of Ca^{2+} dopant on the underlying mechanism of the excellent photocatalytic H_2 generation performance.

2. Experimental section

2.1. Fabrication of CoP_x @CdS

In order to prepare CoP_x @CdS, a certain amount of PPh_3 was first mixing and grinding with commercial CdS. After the self-assembling process of PPh_3 molecules, CoCl_2 with mole ratio of 0.5% were added and annealed at 300–600 °C under argon atmosphere.

2.2. Fabrication of Ca-modified CoP_x @CdS

In detail, the commercial CdS was first mixed with 0.5% mole proportions of Co^{2+} and Ca^{2+} in ethanol solution. After that, the dried mixture was annealed at temperatures at 500 °C under argon atmosphere. The resulting sample was denoted as Ca-modified CoP_x @CdS and applied as photocatalyst for hydrogen generation. For comparison, Ca-modified CdS and CdS-P were prepared by respectively annealing the mixture of CdS/ CaCl_2 and CdS/ PPh_3 under same conditions. The annealing CdS refers to the CdS annealed directly at temperatures at 500 °C under argon atmosphere without any other modification. In addition to Ca^{2+} , CdS has been modified with other metals dopants (Mg^{2+} , Sr^{2+} , Ba^{2+} , Cr^{3+} , Al^{3+} , V^{3+} , Nd^{3+} , and Mo^{5+}) in equal amounts.

2.3. Photocatalytic H_2 evolution activity

Photocatalytic H_2 evolution activities were evaluated under visible-light irradiation. A 300 W xenon arc lamp equipped with a 420 nm cut-off filter was applied as the light source. In all experiments, 260 mL of 0.25 M/0.35 M $\text{Na}_2\text{S}/\text{Na}_2\text{SO}_3$ solution was taken as sacrificial

reductant, and 100 mg of as-prepared sample was used as photocatalyst for achieving H_2 half reaction. The amount of produced H_2 was recorded and analyzed by an online gas chromatograph (GC-8A, Shimadzu Corp., Ar carrier, TCD, Japan) with argon as the carrier gas.

3. Results and discussion

3.1. Fabrication CoP_x @CdS for photocatalytic H_2 production

The detail synthetic routes of CoP_x @CdS photocatalyst have been demonstrated in Fig. 1a. Different from previous reported phosphatizing protocols for synthesize CoP_x -based hybrids, our synthetic strategy based on the organophosphorus compound PPh_3 does not require flammable compounds (e.g. PH_3 or H_2 , etc.) or corrosive conditions, which makes it much safer and easier to manipulate comparing to conventional methods [35–37]. Microstructural characterizations including transmission electron microscopy (TEM), high-resolution TEM (HRTEM) and energy dispersive X-ray spectroscopy (EDX) were utilized to unveil the morphology of CoP_x @CdS structure. As shown in Fig. 1b–e, the amorphous carbon can be observed on the surface of CdS after annealing, where ultrafine CoP_x nanoparticles are separately anchored in the amorphous carbon cages. The elemental mapping displays that C and Co elements uniformly distribute on the surface of CdS, while P element concentrates at interface region between CdS and carbon cages (Fig. S1). In this situation, the amorphous carbon cages act as the substrate for the growth of Co active sites (Fig. S2). The light absorption properties are demonstrated in Fig. 1f, and an increment on light absorption properties over the visible wavelength range has been discovered benefitted from the incorporation of CoP_x [38]. The nanostructure of CdS and CoP_x @CdS has been analyzed by Raman spectroscopy. As shown in Fig. 1g, the first and second longitudinal optical phonon modes (1LO and 2LO) of CdS can be observed [39]. However, the characteristic peaks of CoP_x (262.7 cm^{-1}) cannot be identified due to its low amount when the peak is overlapping with that of CdS [40–42]. Meanwhile, the weak D and G band vibrations of carbon in Fig. S3 suggest that amorphous carbon has been formed [43–45]. Fig. 1h demonstrates the photocatalytic H_2 evolution activities of CoP_x @CdS prepared under calcination temperatures varying from 300 °C to 600 °C. The highest H_2 evolution rate of $1425.2 \mu\text{mol h}^{-1}$ has been realized under the optimized temperature of 500 °C, which indicate the formation of ultrafine CoP_x nanoparticles as well as interfacial interaction between CdS and CoP_x is optimized in this condition as we previously reported [46].

3.2. Synthesis and characterization of Ca-modified CoP_x @CdS

It was known that the long distance between CdS and Co active sites would lead to negative effect on electron transport and insufficient active sites, which is undesirable for photocatalytic reactions. By taking Ca^{2+} dopant to initiate interfacial engineering and to modifying the interface between CdS and CoP_x , the carbon layer would be tightly wrapped on CdS surface (Fig. 2a). It can be regarded that an intimate interfacial contact between CdS and CoP_x has been achieved after Ca^{2+} modification, which may be attributed to ascending reducibility of Ca-modified CdS together with strong interaction between PPh_3 and Ca^{2+} dopant [47]. HRTEM images in Fig. 2b depict that the ultrafine CoP_x nanoparticles are wrapped by thin amorphous carbon layer and anchored on the surface of CdS (the dark/light contrasts observed along radial direction indicate the thickness is approximately 4 nm, Fig. S4). The intimate interfacial contact between CdS and CoP_x may support fast charge transfer with more accessible active sites. The SAED of CdS can be observed in Fig. 2c, which indicates the typical single crystal structure of CdS [48]. The inter-planar spacing $\approx 0.242 \text{ nm}$ is corresponding to the (102) plane of hexagonal CdS, while the d-spacing of 0.28 nm is ascribed to the (002) plane of CoP_x (Fig. 2d). Besides that, EDX mappings (Fig. 2e and Fig. S5) also indicate the uniformly distribution of C, Co, P, and Ca elements on CdS. The chemical compositions of the

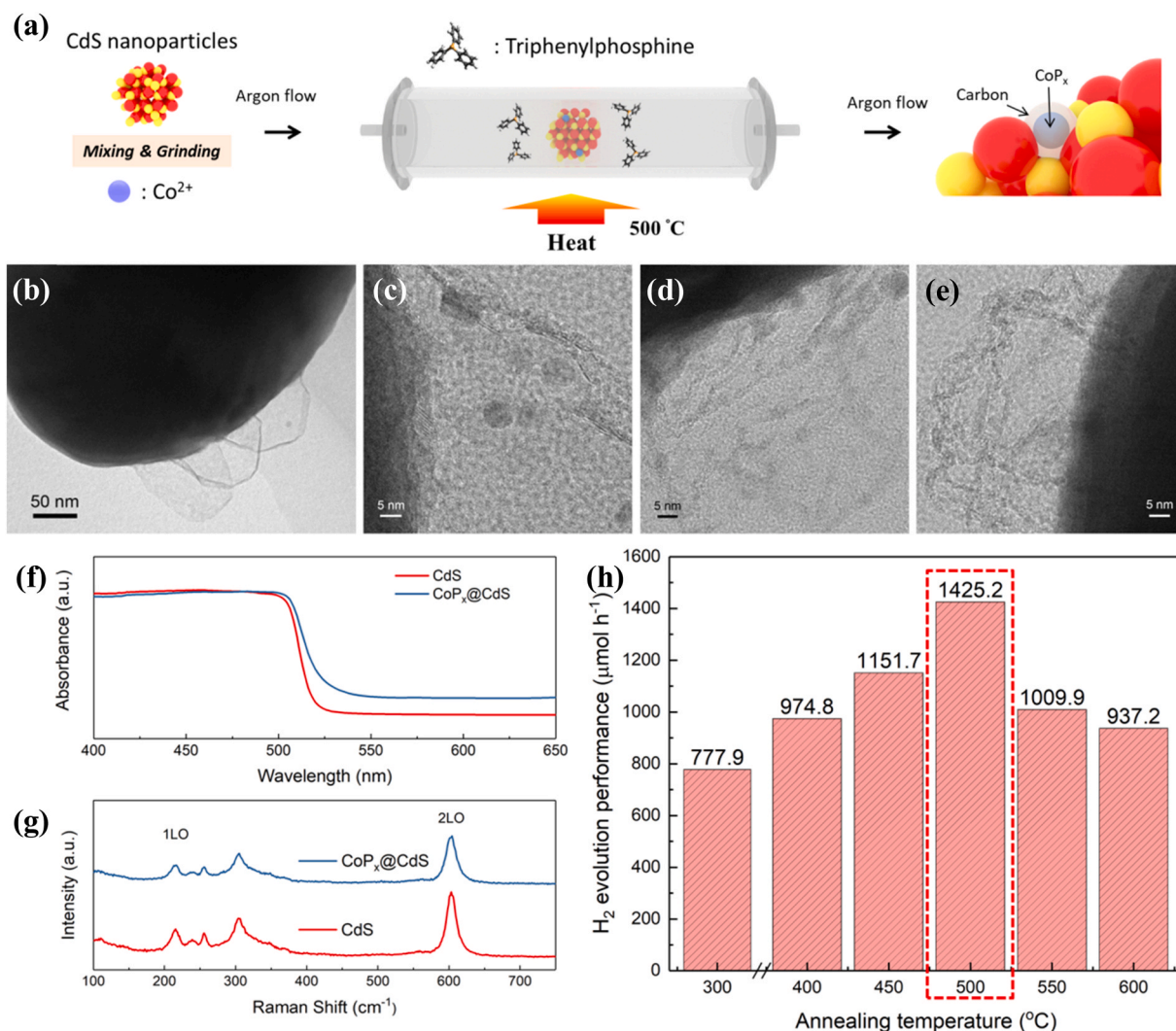


Fig. 1. (a) Schematic illustration on the synthesize processes of $\text{CoP}_x\text{@CdS}$. (b)-(e) TEM images of $\text{CoP}_x\text{@CdS}$. (f) Light absorption spectra and (g) Raman spectra of CdS and Ca-modified $\text{CoP}_x\text{@CdS}$. (h) The H_2 evolution performance of $\text{CoP}_x\text{@CdS}$ annealed under 300–600 °C. (The optimal annealing temperature has been marked in dashed border).

samples (Table S1) were quantitatively determined using inductively coupled plasma-optical emission spectrometry (ICP-OES), and the mass ratio of Ca has been determined to be 0.1 wt%.

X-ray diffraction (XRD) measurements were conducted to investigate the structure of the Ca-modified $\text{CoP}_x\text{@CdS}$ and related samples (Fig. 3a). The XRD results of CdS and Ca-modified $\text{CoP}_x\text{@CdS}$ after calcination under 500 °C show no distinct difference, and the patterns can be indexed into hexagonal CdS (P6₃mc space group, JCPDS No. 06-0314), the peaks at 24.9°, 26.6°, 28.2° are corresponding to (100), (002), and (101) planes of wurtzite CdS, respectively. No obvious shift of the XRD peaks can be observed since the ionic radius of Ca^{2+} (0.099 nm) is similar to that of Cd^{2+} (0.097 nm) (Fig. S6) [49]. Besides, the optical absorption spectra demonstrate the slightly decreased bandgap of Ca-modified CdS comparing to annealed CdS (Fig. S7). In addition, we have conducted X-ray photoelectron spectroscopy (XPS) to study the characteristics of Ca-modified $\text{CoP}_x\text{@CdS}$. Full-survey spectrum shows the existence of Cd, S, C, O, Co, P, and Ca elements in the hybrid (Fig. S8). Meanwhile, the peak centered at 347.3 eV corresponds to $\text{Ca}2p_{3/2}$ indicating that calcium mainly exists as Ca^{2+} . In the high-resolution spectrum of Co 2p (Fig. 3b), the binding energy of 779.6 eV is very close to that of Co $2p_{3/2}$ in CoP_x , which suggests a new electron transfer channel Co-P formed due to interaction between Co and P as previously reported [29,50,51]. Other peaks belong to the Co

$2p_{1/2}$ and Co $2p_{3/2}$ of Co^{2+} components can be attributed into unavoidable surface oxidation. Comparing to the results of EDX (Fig. S9) and ICP-OES, the atomic percentage of Ca^{2+} is much higher when analyzed by XPS (Table S2). Therefore, it can be expected that most of Ca^{2+} ions are existed mainly on the surface of CdS.

The electron transfer process has been elucidated by photoluminescence and time-resolved photoluminescence spectra under room-temperature. In our experiments, a sharp emission band centered at 520 nm and a broad emission band centered at 650 nm can be observed (Fig. 4a) when the photocatalysts were excited at a wavelength of 420 nm. The two characteristic peaks can be assigned to band-to-band transition and surface radiative transition resulting from band-gap emission and trap-state emission, respectively. After Ca doping, the band-gap emission of CdS is dramatically quenched which indicates a low recombination rate of carriers. In contrast, the trap-state emission becomes stronger and broader after Ca doping, this phenomenon reveals that photon-induced carriers can be extracted out to reach the electron trapping sites centered on the surface owing to the increase of Ca-related defects located on the surface [33]. We also notice that both band-gap emission and trap-state emission were remarkably quenched upon the loading of CoP_x . This photoluminescence quench behavior indicates fast charge transfer of electrons from CdS to CoP_x , attributing to higher electron-hole separation ability in Ca-modified $\text{CoP}_x\text{@CdS}$ than that in

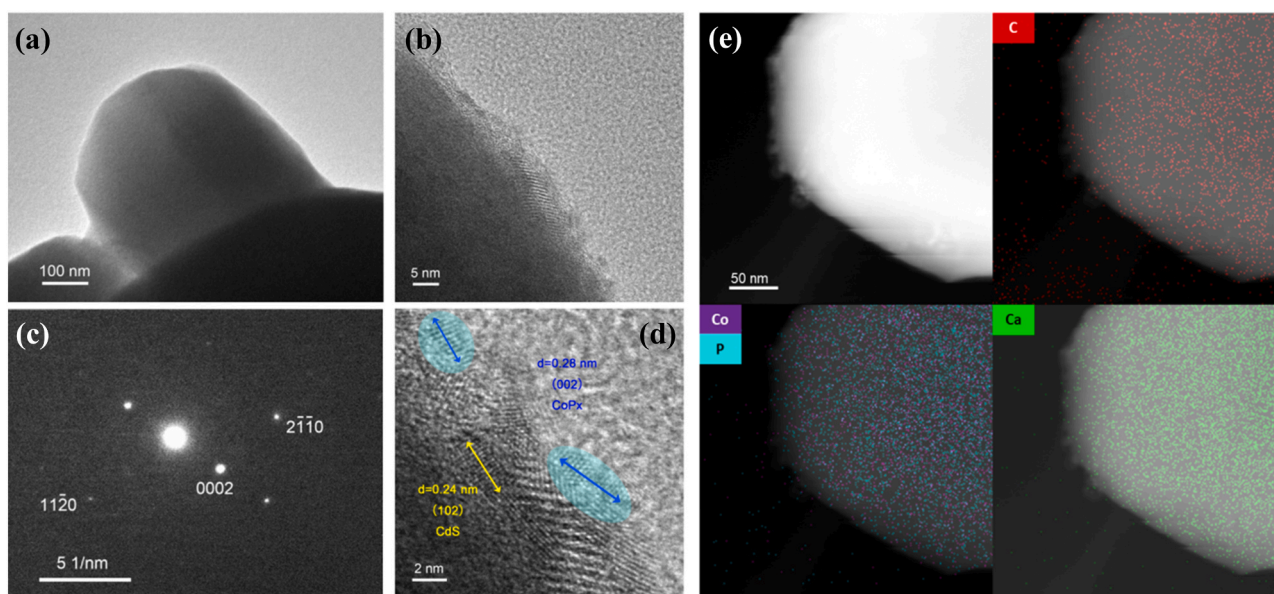


Fig. 2. (a) TEM image, (b) HRTEM image and (c) SAED pattern of Ca-modified CoPx@CdS. (d) HRTEM image on the interface between carbon, CoPx and CdS. (e) HAADF and overlapping EDX mappings of C, Co, P, and C elements on the Ca-modified CoPx@CdS.

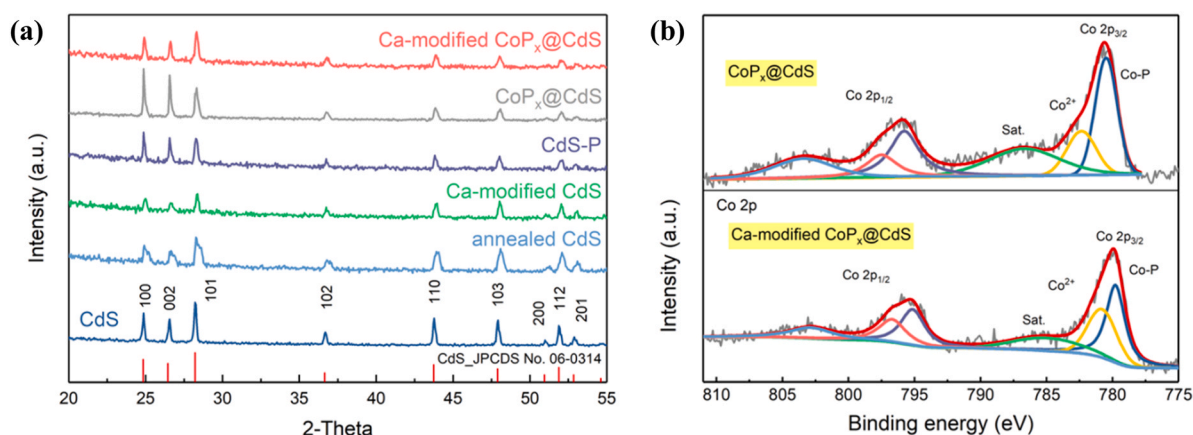


Fig. 3. (a) XRD analysis of CdS, annealed CdS, Ca-modified CdS, CdS-P, CoPx@CdS, and Ca-modified CoPx@CdS. (b) High-resolution XPS spectra of Co 2p for CoPx@CdS and Ca-modified CoPx@CdS.

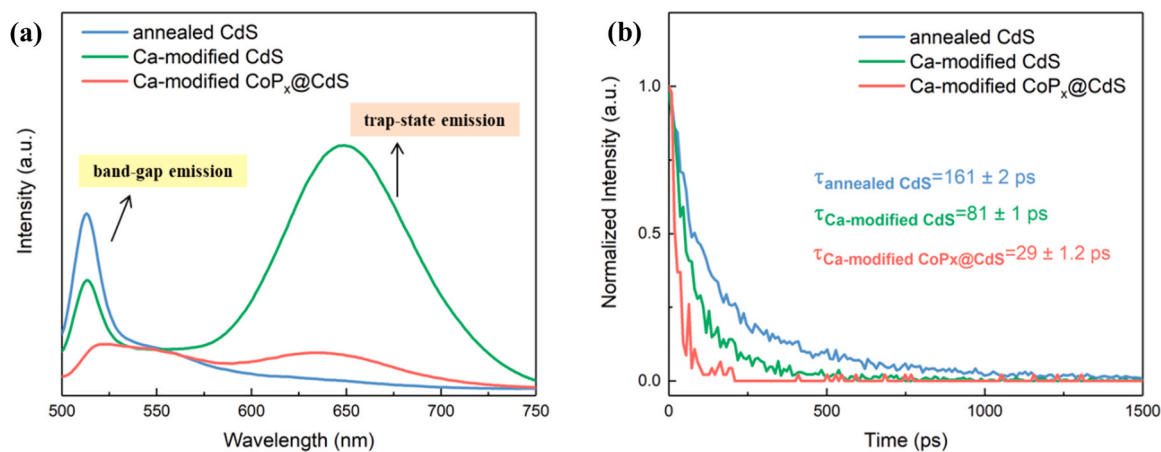


Fig. 4. Charge transfer dynamics analyzed by (a) Steady-state photoluminescence spectra and (b) Time-resolved photoluminescence spectra of annealed CdS, Ca-modified CdS, and Ca-modified CoPx@CdS. (excitation is 420 nm).

pure CdS.

In order to gain in-depth understanding on the transfer dynamics of photogenerated electron in the conduction band of CdS, we have investigated the time-resolved photoluminescence spectra over obtained systems together with the average lifetime of excitons. The charge-transfer dynamics between CdS before and after heat treatment has been studied at first. As shown in Fig. S10, the decay time for CdS (band-gap emission, monitored at 520 nm) has been greatly prolonged from 72 ± 0.7 ps to 161 ± 2 ps after annealing process. This phenomenon can be explained as the photogenerated electrons are surprisingly long lived in the presence of effective trapping states, and endowed with the capability to reach the surface for redox reaction in consequence [30, 52]. In addition, a much shorter carrier lifetime of Ca-modified CdS ($\tau = 81 \pm 1$ ps) compared to annealed CdS ($\tau = 161 \pm 2$ ps) may indicate that Ca dopant is likely to trigger efficient electron extraction from CdS to defect sites (Fig. 4b). Besides that, a dominant time constant has been achieved by $\text{CoP}_x\text{@CdS}$ ($\tau = 37 \pm 1.5$ ps), which features smooth carrier transfer channel with intimate interfacial contact between CdS and CoP_x active species (Fig. S11). The acceleration of decay kinetics for Ca-modified $\text{CoP}_x\text{@CdS}$ further decreases the lifetime from 37 ± 1.5 ps to 29 ± 1.2 ps (Fig. S12). It is believed that fast charge injection from CdS to CoP_x active sites has been realized after Ca^{2+} modification. In this regards, the Ca^{2+} would act as the interfacial mediator for accelerating the electron transfer, attributing to effective electron-hole separation.

3.3. Photocatalytic H_2 generation performance over Ca-modified $\text{CoP}_x\text{@CdS}$

The photocatalytic H_2 evolution performance of as-prepared samples

was evaluated under visible light ($\lambda > 420$ nm) illumination. In detail, CdS alone can only demonstrate a poor H_2 production rate of $284.5 \mu\text{mol h}^{-1}$ (Fig. S13). After annealing, the H_2 production efficiency can be promoted to $639.0 \mu\text{mol h}^{-1}$ (Fig. 5a). Compared with CdS-P ($937.2 \mu\text{mol h}^{-1}$) and $\text{CoP}_x\text{@CdS}$ ($1425.2 \mu\text{mol h}^{-1}$), the photocatalytic H_2 generation rate of Ca-modified $\text{CoP}_x\text{@CdS}$ has reached the highest value of $2441.5 \mu\text{mol h}^{-1}$, with turnover frequency (TOF) as high as 703.2 h^{-1} [53]. The H_2 production performance indicates the obvious enhancement on hydrogen evolution activity of CdS is depending on the interaction between Ca^{2+} and carbon as we have explained in our previous works [46, 54–56]. The advantages of Ca^{2+} for improving the H_2 generation activity over $\text{CoP}_x\text{@CdS}$ have been verified through a comprehensive overview on the photocatalytic H_2 generation performance of $\text{CoP}_x\text{@CdS}$ that modified by other alkaline-earth metal ions (Mg^{2+} , Sr^{2+} , Ba^{2+}) or aliovalent metal ions (such as Cr^{3+} , Al^{3+} , V^{3+} , Nd^{3+} , and Mo^{5+}) as demonstrated in Fig. 5b. The enhancements on H_2 production performance follow in the order of $\text{Ca} > \text{Mo} > \text{Ba} > \text{Mg} > \text{Nd} > \text{Sr} > \text{Al} > \text{V} > \text{Cr}$. The results suggest that alkaline-earth metal ions (Mg^{2+} , Ca^{2+} , Sr^{2+} , Ba^{2+}) play positive roles in modifying CdS for promoting photocatalytic H_2 production.

The apparent quantum efficiency (AQE) and stability measurement of the optimal Ca-modified $\text{CoP}_x\text{@CdS}$ has been provided. The curve of AQE (Fig. 5c) of Ca-modified $\text{CoP}_x\text{@CdS}$ is dependent on the light wavelength which is related to the optical absorbance of the photocatalyst. The performance of Ca-modified $\text{CoP}_x\text{@CdS}$ can reach up to 35.4% at 420 nm and is comparable or even higher than that of previous reported CdS-based photocatalyst systems (Table S3 and Table S4). The cycling stability over Ca-modified $\text{CoP}_x\text{@CdS}$ has been investigated as shown in Fig. 5d. Notably, our photocatalyst exhibits outstanding

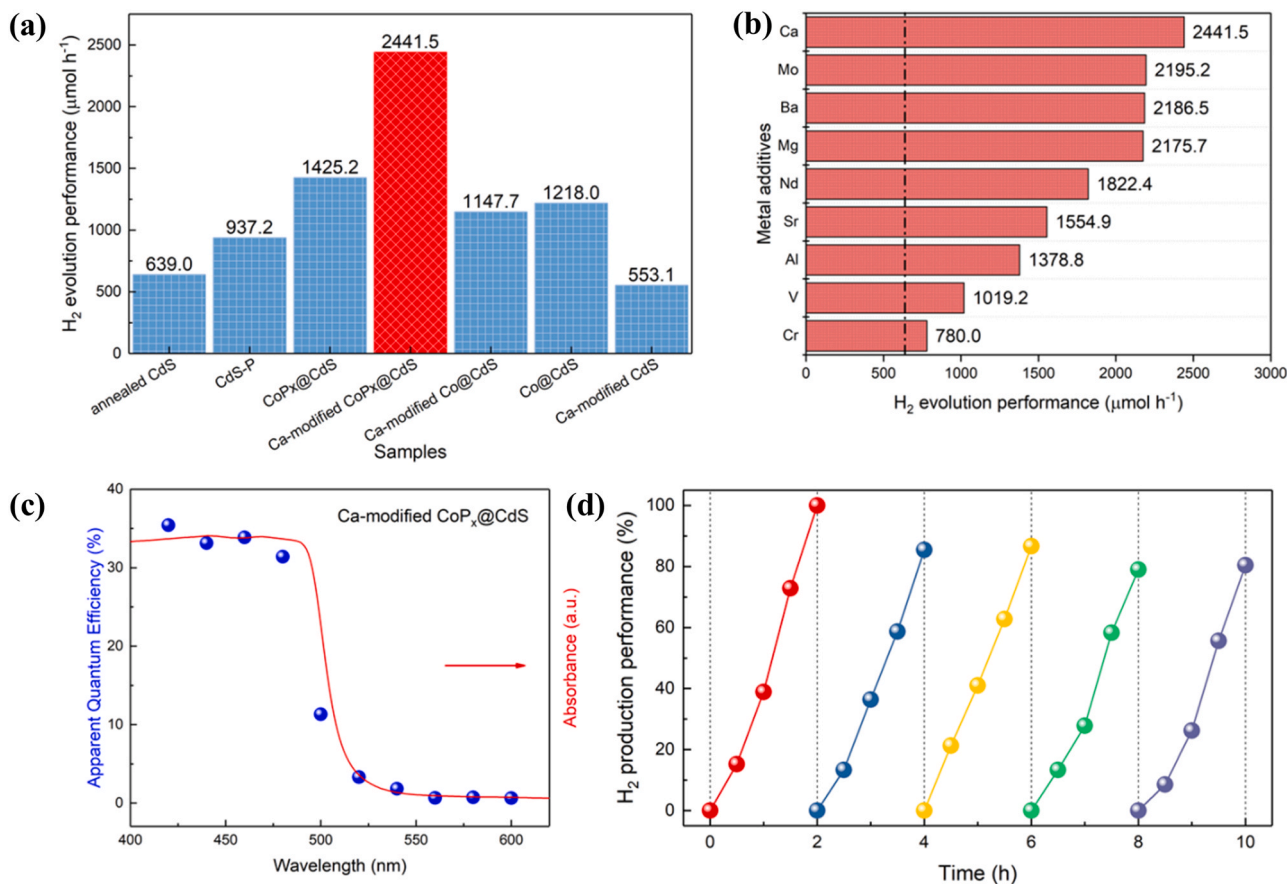


Fig. 5. (a) Comparison on the photocatalytic H_2 evolution performance between annealed CdS, CdS-P, $\text{CoP}_x\text{@CdS}$, Ca-modified $\text{CoP}_x\text{@CdS}$, Ca-modified Co@CdS , Co@CdS and Ca-modified CdS. (b) The photocatalytic H_2 evolution performance of $\text{CoP}_x\text{@CdS}$ modified by different metal dopants. (The dash-dot line represents pristine H_2 generation performance of annealed CdS). (c) AQE curve of Ca-modified $\text{CoP}_x\text{@CdS}$. (d) Cycling stability measurement over Ca-modified $\text{CoP}_x\text{@CdS}$.

stability in $\text{Na}_2\text{S}/\text{Na}_2\text{SO}_3$ solution, and the H_2 evolution performance can retain after five cycling tests without significant decrease. Moreover, the H_2 production of Ca-modified $\text{CoP}_x\text{@CdS}$ depicts linear increasing without apparent decay (Fig. S14), sustaining for 24 h under appropriate conditions (20 mg Ca-modified $\text{CoP}_x\text{@CdS}$ photocatalyst, dispersed into 260 mL of 0.25/0.35 $\text{Na}_2\text{S}/\text{Na}_2\text{SO}_3$ solution, irradiated by 420 nm monochromatic light). A total amount of 289 $\mu\text{mol H}_2$ has been produced with a steady rate of 12 $\mu\text{mol h}^{-1}$ during the course of long-term stability test for 24 h. All these results prove that Ca-modified $\text{CoP}_x\text{@CdS}$ is a robust H_2 production photocatalyst.

3.4. DFT calculation and mechanism analysis

The DFT calculation has been carried out to gain in-depth understanding on the intrinsic properties of Ca-modified $\text{CoP}_x\text{@CdS}$ and underlying H_2 generation mechanism [46]. First of all, the Gibbs free energy change for H^* adsorption (ΔG_{H^*}) values of $\text{CoP}_x\text{@CdS}$ before and after Ca^{2+} modification have been investigated. As shown in Fig. 6a, the obtained ΔG_{H^*} of pristine $\text{CoP}_x\text{@CdS}$ is 0.17 eV, while the ΔG_{H^*} of the system decreases to -0.06 eV when a Cd atom in CdS is replaced by Ca atom. It can be found that Ca^{2+} modification can greatly reduce the $|\Delta G_{\text{H}^*}|$ of $\text{CoP}_x\text{@CdS}$, which signifies Ca-modified $\text{CoP}_x\text{@CdS}$ is kinetic favorable for the adsorption and desorption of hydrogen comparing to that of $\text{CoP}_x\text{@CdS}$. In addition, the charge transferring behavior has been elucidated by analyzing the charge density difference. The charge densities are redistributed at the hetero-interface between CdS and CoP_x layer after Ca^{2+} doping as shown in Fig. S15, which suggest much more

favorable charge transfer between CdS and CoP_x and accumulation of charge on the interface owing to the existence of Ca^{2+} dopant. Moreover, the electron density accumulation and depletion will result in the formation of the built-in electric field between the adjacent CdS and CoP_x layer (Fig. S16 and Table S5).

Similarly, the electron localization functions (Fig. 6b) show that the electron locality of the Co and Cd atoms is weak, while that of the P, S and Ca atoms is strong. This fact can be attributed to that Co can inject electrons into the P atoms and Ca dopants, and Ca^{2+} dopants are bridging the interfacial electronic structure as well as charge transfer between CdS and CoP_x . In addition, the total density of states (TDOS) of $\text{CoP}_x\text{@CdS}$ before and after Ca^{2+} doping have been performed in Fig. S17, pronounced peak changes can be observed in the close vicinity of Fermi level when comparing the DOS between $\text{CoP}_x\text{@CdS}$ and Ca-modified $\text{CoP}_x\text{@CdS}$ systems. The results indicate that electron density near the Fermi level becomes more localized, and an enhancement of local electrical conductivity can be realized owing to the dopant effect of Ca. Meanwhile, to further clarify the effect of Ca doping on the electronic structure, the partial density of states (PDOS) for the orbitals of Cd atoms, Co atoms and Ca atom are shown in Fig. S18. The overlapping of Cd, Co, and Ca orbitals suggest the formation of electron transport channels between Co and Ca which may be the results from orbital hybridization [57,58]. In this regards, photo-excited electrons are first captured in surface defect sites of Ca-modified CdS. Then, the trapped charge carriers would further transfer to CoP_x nanoparticles more easily through the interface of heterostructure due to the lower Fermi level of CoP_x as compared to CdS, and subsequently participate into

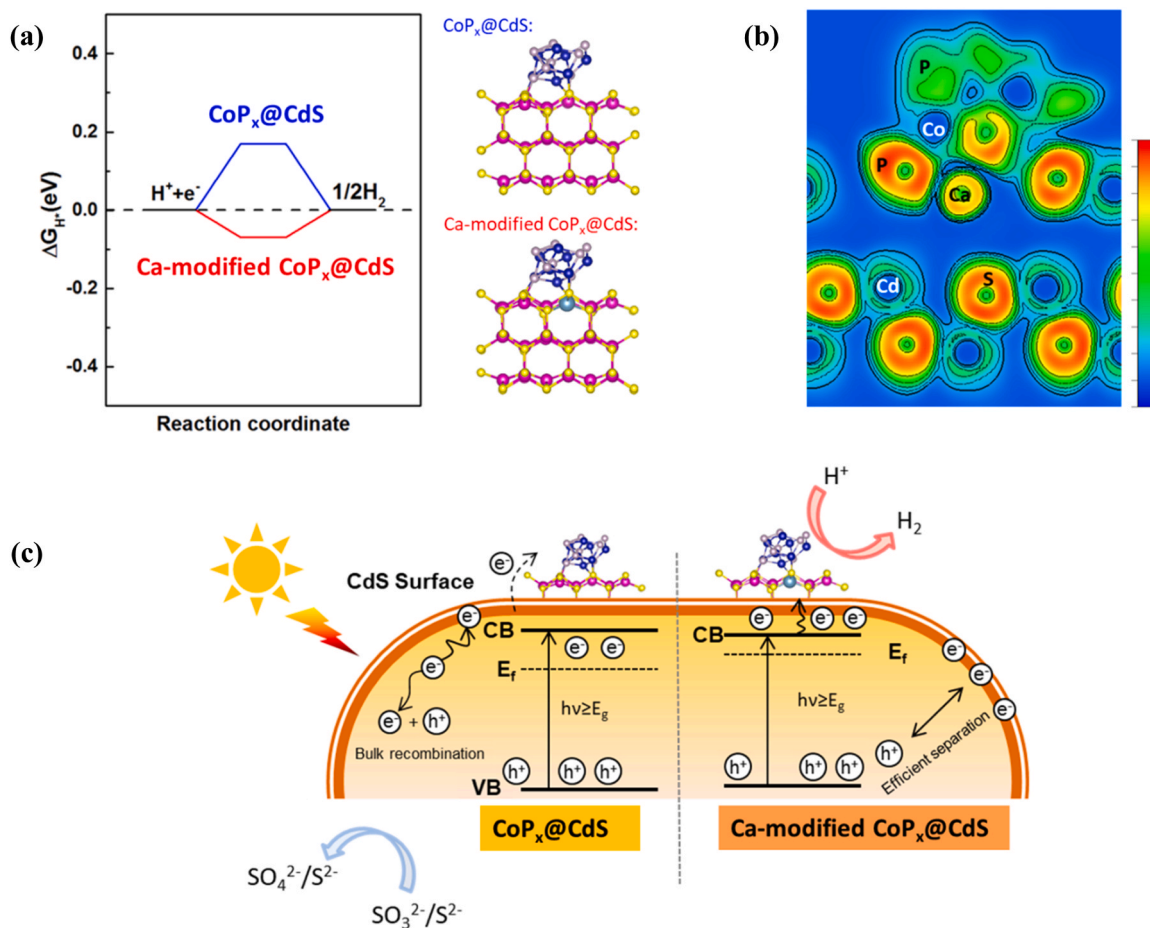


Fig. 6. (a) The calculated Gibbs free energy of H adsorbed $\text{CoP}_x\text{@CdS}$ and Ca-modified $\text{CoP}_x\text{@CdS}$ and corresponding atomic structure. (Yellow balls: Sulfur atoms; Magenta balls: Cadmium atoms; Gray balls: Phosphorus atoms; Blue balls: Cobalt atoms; Wathet balls: Calcium atoms) (b) The side views of the electron localization functions from the (010) section of the Ca-modified $\text{CoP}_x\text{@CdS}$. The concentration of electrons increases from blue to red. (c) Mechanism illustration on charge migration in Ca-modified $\text{CoP}_x\text{@CdS}$. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

photocatalytic H₂ generation [59].

Based on the aforementioned experimental and theoretical results, the critical factors for the remarkable improvement on H₂ evolution performance should be the fast electron transfer behavior as well as efficient charge separation in the hetero-interface of CdS and CoP_x (Fig. 6c). Specifically, the CoP_x@CdS has been endowed with the capability to overcome undesired charge carrier recombination. Meanwhile, the interfacial engineering through Ca²⁺ dopant can promote the thermodynamic kinetics of H₂ production over the surface active sites on CoP_x. Owing to the interplay of CoP_x and Ca²⁺ dopant, the efficient interfacial charge transfer channel between CdS and CoP_x has been formed. Therefore, photogenerated electrons would transfer through the interlayer between CdS and CoP_x more efficiently under light illumination for enhancing H₂ generation.

4. Conclusions

In summary, PPh₃ has been successfully applied to fabricate carbon encapsulated ultrafine CoP_x nanoparticles on the surface of CdS. Meanwhile, Ca²⁺ as an alkaline earth metal ion may easily substitute the Cd²⁺ lattice sites of CdS, which has induced strong trap-state emission together with long-lived charge carriers. In-depth investigations on the transfer dynamics of photogenerated electron demonstrated fast charge transfer behavior in Ca-modified CoP_x@CdS, indicating the enhanced electron-hole separation ability than that of unmodified CdS. The surface Ca²⁺ may act as the interfacial mediator for accelerating the electron transfer from CdS to CoP_x for participate in the H₂ generation reaction. As the result, Ca-modified CoP_x@CdS can achieve 2441.5 μmol h⁻¹ H₂ generation rate with an apparent quantum efficiency of 35.4% at 420 nm. Therefore, we hope this investigation would offer new opportunities and be beneficial for future modification on photocatalysts through metal dopants.

CRedit authorship contribution statement

Xiaohui Ren: Conceptualization, Data curation, Formal analysis, Methodology, Investigation, Writing – original draft. **Fei Liu:** Conceptualization, Investigation, Methodology, Validation. **Qi Wang:** Conceptualization, Validation. **Hui Song:** Investigation, Validation, Writing – review & editing. **Shunqin Luo:** Data curation, Validation, Writing – review & editing. **Sijie Li:** Validation. **Gaoliang Yang:** Visualization. **Bowen Deng:** Validation. **Zongyu Huang:** Software. **Xu-Sheng Wang:** Validation, Resources. **Li Shi:** Conceptualization, Methodology, Writing – review & editing. **Professor Jinhua Ye:** Supervision, Conceptualization, Writing – review & editing, Resources, Validation, Project administration, Funding acquisition.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at doi:10.1016/j.apcatb.2021.120887.

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